

## THERMAL PROPERTIES OF 2D MOLYBDENUM DISULFIDE

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Abstract. The first principle is used to optimize the structure of molybdenum disulfide layers and to calculate the thermal and optical properties. The results include the heat capacity, vibrational entropy, totally electronic energy, Helmholtz free energy and the IR and Raman spectra of molybdenum disulfide layers. The results show that the number of layers of is more important to the thermal properties of molybdenum disulfide than the size of layers.

### 1. Introduction

In recent years, two-dimensional nanomaterials represented by graphene have aroused extensive interest in the field of micro-nano-electronics. People are very concerned about what materials can replace silicon as the basic materials for the next generation of large-scale integrated circuits[1]. Graphene was once the focus of attention, but its bandgap free band structure was not suitable for making circuits. Therefore, molybdenum disulfide ( $\text{MoS}_2$ ), which has similar structural properties but is more excellent in band structure, has gradually become a hot topic in the research of new semiconductor materials[2]. The bulk material  $\text{MoS}_2$  is an indirect bandgap semiconductor with the forbidden band width of 1.29 eV. While monolayer  $\text{MoS}_2$  has a direct bandgap semiconductor with the forbidden band width of 1.8 eV[3]. In the  $\text{MoS}_2$  layer,  $\text{S}^{2-}$  and  $\text{Mo}^{4+}$  are connected with each other by strong covalent bond, while the normal force between  $\text{MoS}_2$  layer and  $\text{MoS}_2$  layer is weak[4]. Therefore, lithium and magnesium plasma can be effectively inserted into the  $\text{MoS}_2$  sheet layer. Therefore,  $\text{MoS}_2$  is widely concerned in the cathode materials of lithium and magnesium ion batteries[5]. Radisavljevic et al. successfully separated the layered  $\text{MoS}_2$  by microcomputer separation method. Moreover, they successfully transplanted the separated  $\text{MoS}_2$  onto the  $\text{SiO}_2/\text{Si}$  substrate[6]. Up to now, there have been a lot of theoretical and experimental studies on the electronic structure and surface properties of  $\text{MoS}_2$  materials, but few studies have been conducted on its optical properties, especially the optical properties of single-layer  $\text{MoS}_2$ . In recent years, first-principles methods based on density functional (DFT) theory have been increasingly used to calculate the optical properties of materials.

In this article, we use the DFT theory compose and cast a shadow under the framework of the plane wave method, using the local density approximation, the single  $\text{MoS}_2$  band structure, density of states and optical properties, such as energy loss spectrum, absorption coefficient and reflectivity, refractive index and extinction coefficient to compare the comprehensive plan, and the results are compared with the existing theoretical results.

## 2. Theoretical details

The present calculations were carried out by VASP code [7, 8], which is a complex package for performing ab-initio quantum-mechanical molecular dynamics (MD) simulations based on first-principles density functional theory (DFT). For calculation of a system, the approach implemented in VASP is based on the (finite-temperature) local-density approximation with the free energy as variable quantity and an exact evaluation of the instantaneous electronic ground state at each MD time step. VASP uses efficient matrix diagonalization schemes and an efficient Pulay/Broyden charge density mixing. These techniques avoid all problems possibly occurring in the original Car-Parrinello method [9], which is based on the simultaneous integration of charged particles, both electronic and ionic, equations of motion. The interaction between them is described by ultra-soft Vanderbilt pseudopotentials (US-PP) [10] or by the projector-augmented wave (PAW) method. US-PP (and the PAW method) allows for a considerable reduction of the number of plane-waves per atom for transition metals and first row elements.

The heat capacity of MoS<sub>2</sub>,  $C_v$ , is represented in terms of contributions of all atoms and possible directions by

$$C_v = \sum_{i,\mu} C_{i,\mu} \quad (1)$$

$$C_{i,\mu} = dk_B \int g_{i,\mu}(\omega) \left( \frac{h\omega}{2k_B T} \right)^2 \frac{\exp\left(\frac{h\omega}{k_B T}\right)}{\left(\exp\left(\frac{h\omega}{k_B T}\right) - 1\right)^2} d\omega \quad (2)$$

In which,  $C_{i,\mu}$  is the contribution of any atom in any possible direction to the heat capacity.  $T$  is temperature in Kelvin,  $k_B$  is Boltzmann constant, and  $h$  is Planck constant. At the low and high temperature limits, we have

$$\lim_{T \rightarrow 0} C_{i,\mu} = 0, \quad \lim_{T \rightarrow \infty} C_{i,\mu} = k_B \quad (3)$$

The Helmholtz free energy of a unit cell,  $A_T$ , is in harmonic approximation

$$A_T = \sum A_{i,\mu} \quad (4)$$

$$A_{i,\mu} = dk_B T \int_0^\infty g_{i,\mu}(\omega) \ln \left( 2 \sin \frac{h\omega}{k_B T} \right) d\omega \quad (5)$$

The entropy of a unit cell,  $S_T$ , is described by

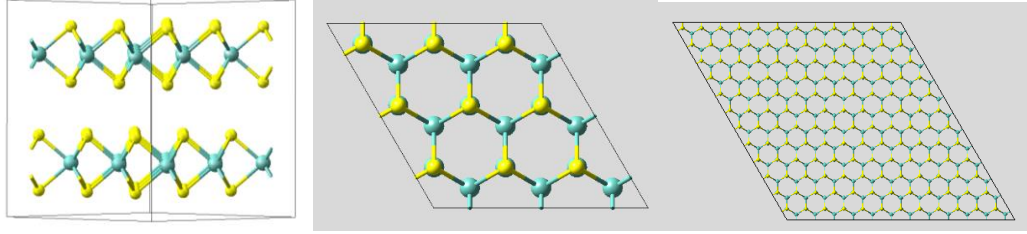
$$S_T = \sum_{i,\mu} S_{i,\mu} \quad (6)$$

$$S_{i,\mu} = dk_B \int_0^\infty g_{i,\mu}(\omega) \left( \frac{h\omega}{2k_B T} \left( \cos \left( \frac{h\omega}{2k_B T} \right) - 1 \right) - \ln \left( 1 - \exp\left(\frac{-h\omega}{k_B T}\right) \right) \right) d\omega \quad (7)$$

Internal energy is the total kinetic energy of a thermodynamic system, which is the energy due to motion of particles, including rotational, translational, and vibrational motion. The potential energy is associated with the vibrational energy of atoms within given molecules or crystals. It includes energy in all of chemical bonds in a system, and the energy of the free electrons or conduction electrons in metals. In this work, the total energy ( $E_T$ ) at temperature  $T$  is the electronic energy of formation plus the vibrational internal energy, i.e.,  $E_T = E_{elec} + ZPE + E_{vib}(T)$ , where  $ZPE$  is the zero point energy.  $E_{vib}(T) = E_T - E_0$  is the change in vibrational internal energy from 0 K. The following relationship holds between the Helmholtz free energy, total energy, and entropy:

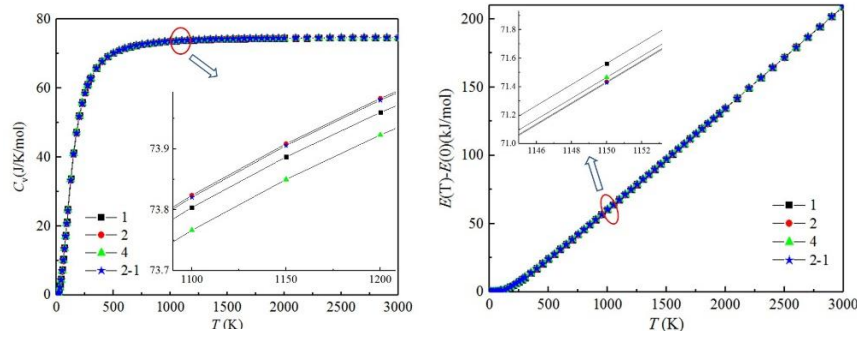
$$A_T = E_T - TS_T \quad (8)$$

In the present simulations, the phonon spectra were calculated on an optimized structure by DFT. All the DFT calculations have been performed in the framework of generalized gradient approximation (GGA) technique considering the exchange and correlation functional as proposed by Perdew-Burke-Ernzerhof (PBE) [11]. There is no magnetic moment in the present model, and the default plane wave cutoff energy is 500 eV. The electronic iterations convergence is  $1.00 \times 10^{-5}$  eV using the blocked Davidson algorithm and reciprocal space projection operators. Fig. 1. is the top view and side view of calculated double layers MoS<sub>2</sub>. In the calculation, we used the first order Methfessel-Paxton smearing with a width of 0.2 eV and the k-spacings of  $0.383 \times 0.383 \times c$  per Angstrom for the monomer(1), double layers(2) and four layers MoS<sub>2</sub>(4). The big size of double layers MoS<sub>2</sub>(2-1) is four times of side length in top view, that is  $4a \times 4b$  when the top views of 1,2,4 are  $a \times b$ .



**Fig. 1.** Views of MoS<sub>2</sub>: side view of 2 (left); top view of 1, 2, 4(middle); top view of 2-1(right)

### 3. Results and discussion



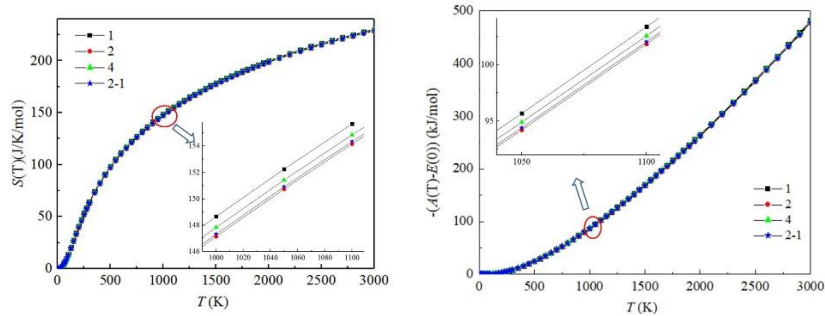
**Fig. 2.** The  $C_v$  curves of the MoS<sub>2</sub> layers **Fig. 3.** The  $E(T)-E(0)$  curves of the MoS<sub>2</sub> layers

Fig.2 shows the results of the  $C_v$  curves of the MoS<sub>2</sub> layers. In the figure, 1, 2, 4 and 2-1 are the monolayer, double layers, four layers and double layers with four cells of MoS<sub>2</sub>, respectively. The  $C_v$  curves in the figure have the same trend. can be described in three stages as temperature changes.  $C_v$  MoS<sub>2</sub> can be described in three stages as temperature changes. Firstly,  $C_v$  of MoS<sub>2</sub> increases rapidly with the increase of temperature when temperature is lower than 500 K. Secondly, at 500 to 1000 K,  $C_v$  of MoS<sub>2</sub> increases slow with temperature. The curves of  $C_v$  are almost parallel to the temperature at the temperature higher than 1000K. In other words,  $C_v$  does not increase with temperature any longer at high temperature. As can be seen in the locally enlarged image,  $C_v$  of monolayer MoS<sub>2</sub> is higher than that of four layers

MoS<sub>2</sub> and lower than that of double layers MoS<sub>2</sub> at the same temperature, such as 1100 K. At the same temperature, the difference  $C_v$  of double layers MoS<sub>2</sub> between one cell and four cells is very small, which is about 0.01J/mol/K.

Fig. 3 shows the  $E(T)-E(0)$  curves of the MoS<sub>2</sub> layers. In this figure, there is a trend chart contrary to  $C_v$  curves.  $E(T)-E(0)$  of MoS<sub>2</sub> layers increase slowly at low temperature while rapidly at high temperature. At the same temperature, such as 1150K, the highest  $E(T)-E(0)$  is for monomer MoS<sub>2</sub> while the lowest is for the double layers. It might be mean that the monomer MoS<sub>2</sub> is less stable than the others in the work and the double layer ones are the most stable MoS<sub>2</sub>. The two kinds of double layers have very close data of  $E(T)-E(0)$ . The lowest  $E(T)-E(0)$  in the figure is for the big cell of double layers MoS<sub>2</sub>. Therefore the number of layers has great influence on  $E(T)-E(0)$  of MoS<sub>2</sub> layers, while the size has little influence on  $E(T)-E(0)$  of MoS<sub>2</sub> layers.

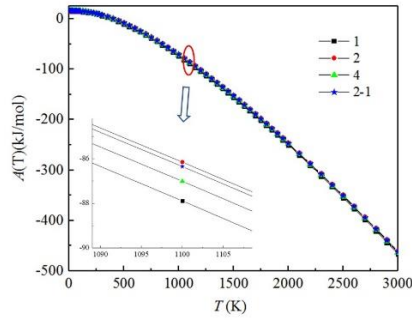
The curves of  $S(T)$  for the MoS<sub>2</sub> layers are shown in Fig.4. As shown in the figure, there is a increase trend for  $S(T)$  of the MoS<sub>2</sub> layers with temperature. curves of  $S(T)$  for the MoS<sub>2</sub> layers are going up slowly at lower temperature than 100 K. It increases quickly when temperature is from 100 ~1000 K. The slope of the curves get smaller and smaller when temperature is higher than 1000 K. In a locally enlarged view, the monomer has highest  $S(T)$ . The double layers MoS<sub>2</sub> has lowest  $S(T)$ . The difference of  $S(T)$  between the two kinds double layers MoS<sub>2</sub> is smaller than that between MoS<sub>2</sub> with different layers. So the number of layers is critical to  $S(T)$  of MoS<sub>2</sub> layers.



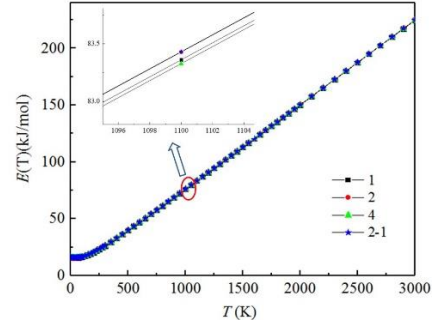
**Fig. 4.** The  $S(T)$  curves of the MoS<sub>2</sub> layers **Fig. 5.** The  $-(A(T)-E(0))$  curves of the MoS<sub>2</sub> layers

The  $-(A(T)-E(0))$  curves of the MoS<sub>2</sub> layers were shown in Fig. 5. In the figure, the  $-(A(T)-E(0))$  of MoS<sub>2</sub> layers increase with temperature. The curves of the MoS<sub>2</sub> layers are flat at low temperature while the slope of curves gets steeper and steeper at the temperature above 250 K. In the local enlarged view,  $-(A(T)-E(0))$  of monomer MoS<sub>2</sub> is highest and that of double layers. The big cell of the double layers has a higher  $-(A(T)-E(0))$  than the small cell double layers MoS<sub>2</sub>. The difference of  $-(A(T)-E(0))$  between the double layers MoS<sub>2</sub> is very small. Therefore size has little effect on  $-(A(T)-E(0))$ , compared with number of layers.

Fig. 6 shows the  $A(T)$  curves of the MoS<sub>2</sub> layers. The figure shows that there is a trend of decrease  $A(T)$  with temperature. The curves of  $A(T)$  are almost flat before 250 K. They began to decrease and speed up to reduce with temperature at higher temperature than 250 K. In the partly enlarged view, the lowest  $A(T)$  is for the monomer MoS<sub>2</sub> in the figure. The double layers MoS<sub>2</sub> have high  $A(T)$  than the fours layers at the same temperature.

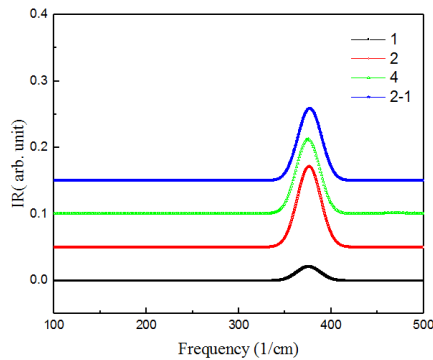


**Fig. 6.** The  $A(T)$  curves of the  $\text{MoS}_2$  layers

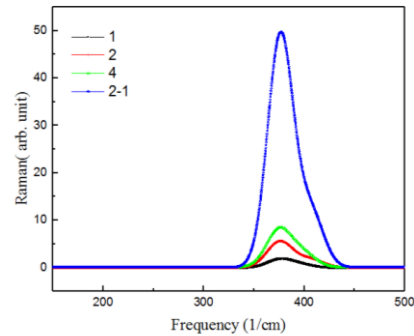


**Fig. 7.** The  $E(T)$  curves of the  $\text{MoS}_2$  layers

$E(T)$  curves of the  $\text{MoS}_2$  layers were shown in Fig. 7. In the figure, the slope of curves is getting more and more negative. In low temperature, the  $E(T)$  of the  $\text{MoS}_2$  layers decrease slightly while accelerate the decline at the temperature above 250 K.  $E(T)$  decrease to lower than -450 kJ/mol. In the partly enlarged view, the monomer  $\text{MoS}_2$  has the lowest  $A(T)$ . The highest  $A(T)$  in the figure is for the double layers  $\text{MoS}_2$ . Two kinds double layers  $\text{MoS}_2$  have similar value in  $A(T)$  and the big cell has the lower  $A(T)$  at the same temperature. The number of layers play a leading role on  $A(T)$  of the  $\text{MoS}_2$  layers.



**Fig. 8.** The IR of the  $\text{MoS}_2$  layers



**Fig. 9.** The Raman of the  $\text{MoS}_2$  layers

Fig.8 describes the IR spectra of the  $\text{MoS}_2$  layers and Fig.9 describes the Raman spectra of the  $\text{MoS}_2$  layers. In the figures, the positions of the peaks are almost identical. In the IR spectra, there is not much difference in peak height except for the low peak of monomer  $\text{MoS}_2$ . In Raman spectra, the difference of peak height was obvious, which low to high order is 1, 2, 4, 2-1. There are shoulder seam marks on the right side of the main peak in the Raman spectra.

#### 4. Conclusions

The less layers of  $\text{MoS}_2$  was simulated by first principle and calculated thermal properties of the layers. The results showed that  $C_v$  of  $\text{MoS}_2$  layers increases with temperature and the slope of the curves is steep at temperature lower than 500 K and tend to be flat at temperature

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higher than 1000 K. The monomer MoS<sub>2</sub> had a higher  $C_v$  than four layers MoS<sub>2</sub> and a lower  $C_v$  than double layers MoS<sub>2</sub>.  $E(T)-E(0)$  and  $E(T)$  increase with temperature. The slope of  $E(T)-E(0)$  and  $E(T)$  curves has a gradually steep trend.  $-(A(T)-E(0))$  curves of MoS<sub>2</sub> layers show a trend of acceleration with temperature. The monomer has the highest  $-(A(T)-E(0))$  at the same temperature.  $A(T)$  curves of MoS<sub>2</sub> layers have countertrend with those of  $-(A(T)-E(0))$ . The  $S(T)$  of MoS<sub>2</sub> layers has an increasing slope at low temperature and a decreasing slope at high temperature. The monomer has the highest  $S(T)$  at the same temperature. In IR and Raman spectra, the MoS<sub>2</sub> layers have almost peak position but different in peak height. The number of layers play more important role in thermal properties of MoS<sub>2</sub> layers than the size does. The double layers MoS<sub>2</sub> is the most stable.

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